

# Synthesis and Crystal Structure of Ternary Sulfides $A_3MS_4$ with $A = \text{Na, Rb}$ and $M = \text{Nb, Ta}$

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The new ternary sulfides  $\text{Na}_3\text{MS}_4$  and  $\text{Rb}_3\text{MS}_4$  ( $M = \text{Nb, Ta}$ ) were prepared from tantalum nitride or niobium powder in the respective alkali metal polysulfide melt. Both sodium sulfides crystallize in form of yellow (Nb) or yellow-orange (Ta) needles in the orthorhombic space group  $Fdd2$  with  $Z = 32$  and  $a = 27.3034(4)$  Å,  $b = 28.4662(1)$  Å, and  $c = 7.9642(1)$  Å,  $R/R_w = 0.024/0.048$  for  $M = \text{Nb}$  and  $a = 27.2322(2)$  Å,  $b = 28.4264(2)$  Å, and  $c = 7.9634(6)$  Å,  $R/R_w = 0.066/0.177$  for  $M = \text{Ta}$ . The rubidium sulfides were obtained as yellow (Nb) and orange (Ta) crystals with irregular shape. They crystallize in the orthorhombic space group  $Pnma$  with  $Z = 4$  and  $a = 9.6109(6)$  Å,  $b = 11.1013(7)$  Å, and  $c = 9.6390(6)$  Å,  $R/R_w = 0.030/0.072$  for  $M = \text{Nb}$  and  $a = 9.6084(5)$  Å,  $b = 11.1067(6)$  Å, and  $c = 9.6426(5)$  Å,  $R/R_w = 0.031/0.070$  for  $M = \text{Ta}$ . The structures of all four compounds are built by stacking isolated  $\text{MS}_3^-$  units with the alkali metal ions. © 1998

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## INTRODUCTION

Ternary sulfides are a fairly well-explored class of compounds. Still, recent years led to a number of new ternary and quaternary alkali metal sulfides and polysulfides of niobium and tantalum (e.g. Refs. (1–8)). The polysulfides were typically prepared from alkali metal polysulfide melts (reactive flux method) at fairly low temperatures of 300–400°C, whereas higher temperatures led to sulfides without sulfur–sulfur bonds.

To our surprise some simple compositions have not yet been investigated. The preparation and the structure of  $\text{K}_3\text{NbS}_4$  were reported;  $\text{K}_3\text{TaS}_4$  crystallizes with an isotopic structure according to its X-ray powder pattern (2).  $\text{CsTaS}_3$  is formed from sulfur and tantalum powders in excess  $\text{Cs}_2\text{S}_5$  melt at 650°C (3). Its crystal structure consists of rods of face-sharing octahedra  $\frac{1}{\infty}[\text{TaS}_{6/2}]$  stacked together in hexagonal rod packing.

Here we report the preparation from polysulfide melts of  $\text{Na}_3\text{MS}_4$  and  $\text{Rb}_3\text{MS}_4$  ( $M = \text{Nb, Ta}$ ) and their structures. The tantalum compounds were initially discovered in

syntheses designed to explore the existence of compounds containing both nitrogen and sulfur. Although we did not identify any of these mixed anion compounds, the title ternaries were discovered.

## EXPERIMENTAL DETAILS

### Synthesis

As many of the reagents and products in this study are moisture-sensitive, the work was carried out in an argon-filled glovebox.

$\text{Ta}_3\text{N}_5$  was obtained from  $\text{TaCl}_5$  powder (Alpha Aesar, 99.99%) by heating in a flow of ammonia (Matheson Gas Products, further dried and stored over sodium) to 750°C. The product is a dark red powder, characterized as single-phase  $\text{Ta}_3\text{N}_5$  (9) by powder X-ray diffraction. The yield is typically quantitative.

TaN can be prepared from  $\text{Ta}_3\text{N}_5$  in a molybdenum crucible when heated inductively (Ameritherm Model Sp-2.5) in nitrogen atmosphere ( $P(\text{N}_2) = 1$  atm) to approximately 1000°C for 20 min. Optical pyrometry was used to monitor the reaction temperature. The resulting black powder was found to be TaN (10) by X-ray diffraction.

$\text{Na}_2\text{S}_2$  and  $\text{Rb}_2\text{S}_3$  were made from the respective alkali metal (Na, Cerac, 99.95%; Rb, Cerac, 99.8%) and sulfur (Cominco American, high purity) in appropriate amounts. After ammonia was condensed onto the metal, the resulting alkali metal solution was added to the sulfur. The mixture was stirred by shaking and then allowed to slowly warm up. The ammonia was evaporated over several hours.  $\text{Na}_2\text{S}_2$  was obtained as a yellow powder. X-ray powder diffraction showed some  $\text{Na}_2\text{S}$  and higher polysulfides as well as the majority phase  $\text{Na}_2\text{S}_2$ , but since the overall alkali metal sulfur ratio had to be 1:1 we used this material for the following reactions.  $\text{Rb}_2\text{S}_3$  is an orange-yellow powder, which was X-ray single phase as determined by X-ray powder diffraction.

$\text{Na}_3\text{TaS}_4$  was prepared from two different reaction mixtures: (i)  $\text{Na}_2\text{S}$  powder (Cerac, 99.9%), elemental sulfur powder, and  $\text{Ta}_3\text{N}_5$  and (ii)  $\text{Na}_2\text{S}$ , elemental sulfur, and

TaN. Both mixtures corresponded to a Na/Ta/S molar ratio of approximately 10:1:10. The reactants were sealed in evacuated silica tubes, heated to 800°C within 24 h, kept at 800°C for 100 h, and slowly cooled down to room temperature within 100 h. After the treatment the products in both tubes looked like solidified red melts. Two kinds of crystals were visually observed under a microscope in the raw powder: yellow hexagonal plates (excess  $\text{Na}_2\text{S}_2$ ) and yellow-orange needles. The sample was then washed with dry DMF under inert atmosphere to remove excess sodium sulfides.

$\text{Na}_3\text{NbS}_4$  can be prepared in an analogous fashion from  $\text{Na}_2\text{S}$ , sulfur, and niobium powder (Na:Nb:S = 10:1:10).  $\text{Rb}_3\text{NbS}_4$  was prepared from excess  $\text{Rb}_2\text{S}_3$  and niobium powder and  $\text{Rb}_3\text{TaS}_4$  from  $\text{Rb}_2\text{S}_3$  and  $\text{Ta}_3\text{N}_5$  (Rb:M:S = 5:1:7.5).  $\text{Na}_3\text{NbS}_4$  crystallizes as yellow needles,  $\text{Rb}_3\text{NbS}_4$  (yellow) and  $\text{Rb}_3\text{TaS}_4$  (orange) as crystals with bulky shape.

Crystals of each compound were manually selected and sealed in glass capillaries for single-crystal X-ray studies. All products are moisture-sensitive.

After the  $A_3MS_4$  stoichiometry was established from the structure solutions, preparation of single-phase  $\text{Na}_3\text{TaS}_4$  samples was attempted. For that purpose mixtures of  $\text{Ta}_3\text{N}_5$  with  $\text{Na}_2\text{S}_2$  powder and elemental sulfur and of

TaN,  $\text{Na}_2\text{S}_2$ , and elemental sulfur at the molar ratio of 3:1:4 (Na:Ta:S) were prepared. The two reaction mixtures were pressed into pellets, placed separately in aluminum oxide crucibles in sealed evacuated silica tubes, and heated to 800°C (from room temperature to 800°C within 50 h, at 800°C for 100 h, slow cooling over a period of 100 h). Both pellets melted in the course of the reactions. Products in both tubes had a black coating on the surface, but they were beige powders in the bulk. Powder X-ray diffraction data were collected and analyzed for both samples. The weight changes of these mixtures during the reactions were not established, but we suspect the formation of  $\text{N}_2$  in a redox reaction with sulfur or polysulfide ions, respectively. Since no other solid product was found anywhere in the tubes and the products appeared to be single phase, the yields are approximately quantitative. Analogous experiments to prepare single phase samples of the three other compositions were not carried out.

#### EDX Analysis of $\text{Na}_3\text{TaS}_4$

To establish the approximate composition of the yellow-orange needles of the sodium tantalum sulfide, EDX analysis of several such crystals was carried out on a scanning electron microscope (JEOL 733). Measurements yielded an

TABLE 1  
Crystallographic Data Concerning the Compounds  $A_3MS_4$  ( $A = \text{Na, Rb}; M = \text{Nb, Ta}$ )

Compound	$\text{Na}_3\text{NbS}_4$	$\text{Na}_3\text{TaS}_4$	$\text{Rb}_3\text{NbS}_4$	$\text{Rb}_3\text{TaS}_4$
Crystal shape	Needle	Needle	Bulk	Bulk
Dimensions/mm	$0.08 \times 0.08 \times 0.2$	$0.08 \times 0.1 \times 0.33$	$0.1 \times 0.1 \times 0.1$	$0.15 \times 0.15 \times 0.15$
Space group, $Z$	$Fdd2$ (No. 43), 32		$Pnma$ (No. 62), 4	
$a$ , Å	27.3034(4)	27.2322(2)	9.6109(6)	9.6084(5)
$b$ , Å	28.4662(1)	28.4264(2)	11.1013(7)	11.1067(6)
$c$ , Å	7.9642(1)	7.9634(6)	9.6390(6)	9.6426(5)
Cell volume, Å <sup>3</sup>	6190.0(2)	6164.6(2)	1028.4(2)	1029.0(1)
Calculated density/g/cm <sup>3</sup>	2.49	3.26	3.08	3.65
Radiation	MoK $\alpha$			
Monochromator	Graphite			
Absorption correction	SADABS (25)			
Min. transmission	0.638	0.393	0.375	0.445
$\mu/\text{mm}^{-1}$	2.69	15.40	15.99	25.49
$2\theta_{\text{max}}/\text{deg}$	56.7	52.8	55.6	55.7
$h, k, l$	–36 to 35	–23 to 33	–12 to 11	–11 to 12
	–29 to 37	–33 to 35	–14 to 10	–14 to 14
	–10 to 10	–9 to 7	–9 to 12	–12 to 9
Independent reflections	3288	2718	1211	1193
Reflections with $F_0^2 \geq 4\sigma F_0^2$	3020	2630	1031	1080
Variables	147	146	44	44
Goodness of fit (goof)	0.989	1.160	1.028	1.223
$R_{\text{int}}$	0.038	0.051	0.048	0.046
$R/R_w$	0.024/0.048	0.066/0.177	0.030/0.072	0.031/0.070
Extinction parameter	0.00028(1)	—	0.0217(9)	0.0050(3)
Largest difference peak	0.67	6.68	1.01	1.84
Flack parameter	0.02(4)	0.05(3)	—	—

TABLE 2  
Atomic Coordinates for Na<sub>3</sub>MS<sub>4</sub> (M = Nb, Ta)<sup>a</sup>

Atom	Site	x	y	z	U <sub>iso</sub> (Å <sup>2</sup> )
Ta(1)	16b	<b>0.00973(3)</b>	<b>0.11555(3)</b>	<b>0.5528(1)</b>	<b>0.0126(3)</b>
		0.24027	0.13445	0.4472	
Nb(1)		<i>0.24010(1)</i>	<i>0.13465(1)</i>	<i>0.44859(4)</i>	<i>0.01379(9)</i>
Ta(2)	16b	<b>0.12245(4)</b>	<b>0.00614(3)</b>	<b>0.0484(1)</b>	<b>0.0139(3)</b>
		0.12755	0.24386	0.9516	
Nb(1)		<i>0.12737(1)</i>	<i>0.24385(1)</i>	<i>0.95322(4)</i>	<i>0.01511(9)</i>
S(1)	16b	<b>0.2491(2)</b>	<b>0.0937(2)</b>	<b>0.562(1)</b>	<b>0.020(1)</b>
		0.0009	0.1563	0.438	
		<i>0.00085(4)</i>	<i>0.15611(4)</i>	<i>0.4400(2)</i>	<i>0.0259(2)</i>
S(2)	16b	<b>0.1864(3)</b>	<b>0.1563(2)</b>	<b>0.9129(9)</b>	<b>0.019(1)</b>
		0.0637	0.0937	0.0871	
		<i>0.06387(4)</i>	<i>0.09382(4)</i>	<i>0.0864(1)</i>	<i>0.0232(2)</i>
S(3)	16b	<b>0.0840(3)</b>	<b>0.2477(2)</b>	<b>0.056(1)</b>	<b>0.025(1)</b>
		0.1661	0.0023	0.944	
		<i>0.16596(4)</i>	<i>0.00226(4)</i>	<i>0.9463(2)</i>	<i>0.0255(2)</i>
S(4)	16b	<b>0.1556(3)</b>	<b>0.1859(2)</b>	<b>0.4000(9)</b>	<b>0.023(2)</b>
		0.0944	0.0641	0.6000	
		<i>0.09485(4)</i>	<i>0.06423(4)</i>	<i>0.6030(2)</i>	<i>0.0240(2)</i>
S(5)	16b	<b>0.0544(3)</b>	<b>0.0503(2)</b>	<b>0.4973(9)</b>	<b>0.021(1)</b>
		0.1956	0.1997	0.5027	
		<i>0.19527(4)</i>	<i>0.20005(4)</i>	<i>0.5052(1)</i>	<i>0.0224(2)</i>
S(6)	16b	<b>0.0500(3)</b>	<b>0.1594(2)</b>	<b>0.7483(9)</b>	<b>0.021(1)</b>
		0.2000	0.0906	0.2517	
		<i>0.19999(4)</i>	<i>0.09083(3)</i>	<i>0.2530(1)</i>	<i>0.0212(2)</i>
S(7)	16b	<b>0.1722(3)</b>	<b>0.0381(2)</b>	<b>0.2480(9)</b>	<b>0.024(2)</b>
		0.0778	0.2119	0.7512	
		<i>0.07734(4)</i>	<i>0.21199(4)</i>	<i>0.7540(1)</i>	<i>0.0249(3)</i>
S(8)	16b	<b>0.0566(3)</b>	<b>0.0531(2)</b>	<b>0.9997(9)</b>	<b>0.022(1)</b>
		0.1934	0.1969	0.0003	
		<i>0.19333(4)</i>	<i>0.19697(4)</i>	<i>0.0021(1)</i>	<i>0.0238(2)</i>
Na(1)	16b	<b>0.0953(6)</b>	<b>0.1375(6)</b>	<b>0.150(3)</b>	<b>0.070(6)</b>
		0.1547	0.1125	0.850	
		<i>0.15426(7)</i>	<i>0.11237(9)</i>	<i>0.8531(4)</i>	<i>0.0681(8)</i>
Na(2)	16b	<b>0.1368(6)</b>	<b>0.0922(5)</b>	<b>0.676(3)</b>	<b>0.052(4)</b>
		0.1132	0.1578	0.324	
		<i>0.11229(8)</i>	<i>0.15697(7)</i>	<i>0.3283(3)</i>	<i>0.0544(6)</i>
Na(3)	16b	<b>0.1297(4)</b>	<b>0.2219(4)</b>	<b>0.728(2)</b>	<b>0.022(2)</b>
		0.1203	0.0281	0.272	
		<i>0.12012(6)</i>	<i>0.02762(6)</i>	<i>0.2718(2)</i>	<i>0.0255(4)</i>
Na(4)	16b	<b>0.0022(5)</b>	<b>0.2463(4)</b>	<b>0.766(2)</b>	<b>0.034(3)</b>
		0.2478	0.0037	0.234	
		<i>0.24785(7)</i>	<i>0.00362(6)</i>	<i>0.2317(3)</i>	<i>0.0395(5)</i>
Na(5)	16b	<b>0.2261(5)</b>	<b>0.1264(4)</b>	<b>0.236(2)</b>	<b>0.026(2)</b>
		0.0239	0.1236	0.764	
		<i>0.02433(6)</i>	<i>0.12300(6)</i>	<i>0.7668(2)</i>	<i>0.0261(4)</i>
Na(6)	8a	$\frac{1}{4}$	$\frac{1}{4}$	<b>0.000(2)</b>	<b>0.038(5)</b>
		0	0	0.000	
		0	0	<i>0.0000(4)</i>	<i>0.0391(7)</i>
Na(7)	8a	$\frac{1}{4}$	$\frac{1}{4}$	<b>0.488(2)</b>	<b>0.035(4)</b>
		0	0	0.512	
		0	0	<i>0.5122(4)</i>	<i>0.0396(7)</i>

<sup>a</sup> Boldface type refers to Na<sub>3</sub>TaS<sub>4</sub>; lightface roman type refers to Na<sub>3</sub>TaS<sub>4</sub> data transformed to the other enantiomeric form; italic refers to Na<sub>3</sub>NbS<sub>4</sub>.

average stoichiometry of Na<sub>2.3</sub>TaS<sub>4.0</sub>. As partial loss of sodium in our experience sometimes takes place in high vacuum during analysis of crystal surfaces, and since the error on the sodium content should be the largest and this is a standardless determination, these results are consistent with the ideal Na<sub>3</sub>TaS<sub>4</sub> stoichiometry.

### Crystal Structure Determination

Selected crystals were first tested on a precession camera. Well-diffracting crystals showing sharp reflections were then studied on a CCD diffractometer (Bruker SMART) at room temperature. Yellow (Na<sub>3</sub>NbS<sub>4</sub>) and yellow-orange (Na<sub>3</sub>TaS<sub>4</sub>) needles were indexed with an F-centered orthorhombic unit cell. Intensity data were taken for each crystal in a hemisphere scan mode with 1321 frames (886 for Na<sub>3</sub>TaS<sub>4</sub>) taken via  $\omega$  rotation ( $\Delta\omega = 0.3^\circ$ ) and 30 s per frame. Systematic extinctions were consistent with space groups *Fddd* and *Fdd2*. Structure solution using direct

TABLE 3  
Atomic Thermal Displacement Parameters (Å<sup>2</sup>) for Na<sub>3</sub>MS<sub>4</sub> (M = Nb, Ta)<sup>a</sup>

Atom	U <sub>11</sub>	U <sub>22</sub>	U <sub>33</sub>	U <sub>23</sub>	U <sub>13</sub>	U <sub>12</sub>
Ta(1)	<b>0.0195(5)</b>	<b>0.0089(4)</b>	<b>0.0096(5)</b>	<b>0.0005(4)</b>	<b>-0.0006(5)</b>	<b>-0.0006(3)</b>
Nb(1)	<i>0.0127(2)</i>	<i>0.0147(2)</i>	<i>0.0139(2)</i>	<i>-0.0005(2)</i>	<i>-0.0004(2)</i>	<i>-0.0003(1)</i>
Ta(2)	<b>0.0205(5)</b>	<b>0.0118(4)</b>	<b>0.0096(5)</b>	<b>-0.0003(4)</b>	<b>-0.0009(5)</b>	<b>-0.0022(3)</b>
Nb(2)	<i>0.0133(2)</i>	<i>0.0176(2)</i>	<i>0.0144(2)</i>	<i>-0.0007(2)</i>	<i>0.0000(2)</i>	<i>-0.0016(1)</i>
S(1)	<b>0.031(3)</b>	<b>0.020(3)</b>	<b>0.009(3)</b>	<b>-0.004(3)</b>	<b>-0.003(3)</b>	<b>0.004(2)</b>
	<i>0.0318(6)</i>	<i>0.0257(5)</i>	<i>0.0201(6)</i>	<i>-0.0053(5)</i>	<i>-0.0037(5)</i>	<i>-0.0016(4)</i>
S(2)	<b>0.022(3)</b>	<b>0.022(3)</b>	<b>0.014(3)</b>	<b>0.000(2)</b>	<b>0.002(3)</b>	<b>0.007(3)</b>
	<i>0.0150(5)</i>	<i>0.0324(6)</i>	<i>0.0223(6)</i>	<i>0.0018(5)</i>	<i>0.0010(4)</i>	<i>0.0060(4)</i>
S(3)	<b>0.030(3)</b>	<b>0.032(3)</b>	<b>0.015(4)</b>	<b>-0.001(3)</b>	<b>-0.007(3)</b>	<b>-0.005(3)</b>
	<i>0.0236(5)</i>	<i>0.0341(6)</i>	<i>0.0188(6)</i>	<i>-0.0036(5)</i>	<i>-0.0032(5)</i>	<i>-0.0005(4)</i>
S(4)	<b>0.036(4)</b>	<b>0.014(3)</b>	<b>0.019(4)</b>	<b>0.003(2)</b>	<b>-0.002(3)</b>	<b>0.010(3)</b>
	<i>0.0274(6)</i>	<i>0.0210(6)</i>	<i>0.0237(6)</i>	<i>-0.0003(4)</i>	<i>0.0008(5)</i>	<i>0.0064(4)</i>
S(5)	<b>0.028(4)</b>	<b>0.018(3)</b>	<b>0.017(4)</b>	<b>-0.002(2)</b>	<b>0.000(3)</b>	<b>0.004(3)</b>
	<i>0.0228(5)</i>	<i>0.0212(5)</i>	<i>0.0232(6)</i>	<i>-0.0040(4)</i>	<i>-0.0007(4)</i>	<i>0.0054(4)</i>
S(6)	<b>0.027(3)</b>	<b>0.018(3)</b>	<b>0.019(4)</b>	<b>-0.007(2)</b>	<b>0.001(3)</b>	<b>-0.006(3)</b>
	<i>0.0195(5)</i>	<i>0.0208(5)</i>	<i>0.0232(6)</i>	<i>-0.0040(4)</i>	<i>-0.0020(4)</i>	<i>-0.0045(4)</i>
S(7)	<b>0.032(4)</b>	<b>0.028(3)</b>	<b>0.012(4)</b>	<b>-0.005(3)</b>	<b>0.001(3)</b>	<b>-0.013(3)</b>
	<i>0.0242(5)</i>	<i>0.0294(6)</i>	<i>0.0211(6)</i>	<i>-0.0031(5)</i>	<i>-0.0024(4)</i>	<i>-0.0096(4)</i>
S(8)	<b>0.035(4)</b>	<b>0.019(3)</b>	<b>0.011(4)</b>	<b>0.001(2)</b>	<b>0.002(3)</b>	<b>0.005(3)</b>
	<i>0.0216(5)</i>	<i>0.0254(5)</i>	<i>0.0243(6)</i>	<i>0.0005(5)</i>	<i>-0.0022(4)</i>	<i>0.0053(4)</i>
Na(1)	<b>0.041(9)</b>	<b>0.07(1)</b>	<b>0.104(17)</b>	<b>-0.05(1)</b>	<b>-0.001(9)</b>	<b>-0.015(8)</b>
	<i>0.033(1)</i>	<i>0.079(2)</i>	<i>0.093(2)</i>	<i>-0.054(2)</i>	<i>0.007(1)</i>	<i>-0.013(1)</i>
Na(2)	<b>0.056(9)</b>	<b>0.033(7)</b>	<b>0.07(1)</b>	<b>-0.017(7)</b>	<b>-0.025(8)</b>	<b>-0.001(7)</b>
	<i>0.038(1)</i>	<i>0.042(1)</i>	<i>0.083(2)</i>	<i>-0.015(1)</i>	<i>-0.027(1)</i>	<i>0.0016(9)</i>
Na(3)	<b>0.024(6)</b>	<b>0.023(5)</b>	<b>0.020(6)</b>	<b>-0.002(4)</b>	<b>-0.003(4)</b>	<b>-0.002(4)</b>
	<i>0.0251(9)</i>	<i>0.0254(9)</i>	<i>0.026(1)</i>	<i>0.0022(7)</i>	<i>0.0010(7)</i>	<i>-0.0035(7)</i>
Na(4)	<b>0.038(6)</b>	<b>0.025(5)</b>	<b>0.04(1)</b>	<b>0.000(5)</b>	<b>-0.003(6)</b>	<b>0.002(5)</b>
	<i>0.0265(9)</i>	<i>0.0300(9)</i>	<i>0.062(2)</i>	<i>0.0044(10)</i>	<i>-0.002(1)</i>	<i>0.0033(7)</i>
Na(5)	<b>0.030(6)</b>	<b>0.032(6)</b>	<b>0.017(6)</b>	<b>0.002(5)</b>	<b>-0.002(5)</b>	<b>0.002(5)</b>
	<i>0.0251(9)</i>	<i>0.0318(9)</i>	<i>0.0214(9)</i>	<i>0.0017(7)</i>	<i>0.0015(7)</i>	<i>0.0002(7)</i>
Na(6)	<b>0.06(1)</b>	<b>0.033(9)</b>	<b>0.02(1)</b>	<b>0</b>	<b>0</b>	<b>0.014(9)</b>
	<i>0.056(2)</i>	<i>0.036(2)</i>	<i>0.026(2)</i>	<i>0</i>	<i>0</i>	<i>0.017(1)</i>
Na(7)	<b>0.05(1)</b>	<b>0.05(1)</b>	<b>0.007(9)</b>	<b>0</b>	<b>0</b>	<b>0.015(9)</b>
	<i>0.032(2)</i>	<i>0.062(2)</i>	<i>0.025(2)</i>	<i>0</i>	<i>0</i>	<i>0.017(1)</i>

<sup>a</sup> Boldface type refers to Na<sub>3</sub>TaS<sub>4</sub>, and italic type refers to Na<sub>3</sub>NbS<sub>4</sub>.

TABLE 4  
Atomic Coordinates for  $\text{Rb}_3\text{MS}_4$  ( $M = \text{Nb, Ta}$ )<sup>a</sup>

Atom	Site	x	y	z	$U_{\text{iso}}$ ( $\text{\AA}^2$ )
<b>Ta(1)</b>	<b>4c</b>	<b>0.28174(4)</b>	$\frac{1}{4}$	<b>0.01461(4)</b>	<b>0.0136(2)</b>
<i>Nb(1)</i>		<i>0.28145(5)</i>	$\frac{1}{4}$	<i>0.01456(5)</i>	<i>0.0154(2)</i>
<b>Rb(1)</b>	<b>8d</b>	<b>0.05029(8)</b>	<b>0.04291(7)</b>	<b>0.28959(8)</b>	<b>0.0244(2)</b>
		<i>0.05018(5)</i>	<i>0.04318(4)</i>	<i>0.29011(5)</i>	<i>0.0267(2)</i>
<b>Rb(2)</b>	<b>4c</b>	<b>0.3544(1)</b>	$\frac{1}{4}$	<b>0.5840(1)</b>	<b>0.0382(3)</b>
		<i>0.35454(7)</i>	$\frac{1}{4}$	<i>0.58452(7)</i>	<i>0.0401(2)</i>
<b>S(1)</b>	<b>8d</b>	<b>0.1678(2)</b>	<b>0.4193(2)</b>	<b>0.9481(2)</b>	<b>0.0301(5)</b>
		<i>0.1678(1)</i>	<i>0.4196(1)</i>	<i>0.9482(1)</i>	<i>0.0330(3)</i>
<b>S(2)</b>	<b>8d</b>	<b>0.0003(3)</b>	$\frac{1}{4}$	<b>0.5781(3)</b>	<b>0.0217(5)</b>
		<i>-0.0002(2)</i>	$\frac{1}{4}$	<i>0.5780(2)</i>	<i>0.0247(4)</i>
<b>S(3)</b>	<b>8d</b>	<b>0.2992(3)</b>	$\frac{1}{4}$	<b>0.2488(3)</b>	<b>0.0207(5)</b>
		<i>0.2990(2)</i>	$\frac{1}{4}$	<i>0.2496(2)</i>	<i>0.0228(4)</i>

<sup>a</sup> Boldface type refers to  $\text{Rb}_3\text{TaS}_4$ ; italic type refers to  $\text{Rb}_3\text{NbS}_4$ .

TABLE 5  
Atomic Thermal Displacement Parameters ( $\text{\AA}^2$ ) for  $\text{Rb}_3\text{MS}_4$  ( $M = \text{Nb, Ta}$ )<sup>a</sup>

Atom	$U_{11}$	$U_{22}$	$U_{33}$	$U_{23}$	$U_{13}$	$U_{12}$
<b>Ta(1)</b>	<b>0.0112(2)</b>	<b>0.0183(2)</b>	<b>0.0113(2)</b>	<b>0</b>	<b>-0.0005(1)</b>	<b>0</b>
<i>Nb(1)</i>	<i>0.0153(3)</i>	<i>0.0165(3)</i>	<i>0.0145(3)</i>	<i>0</i>	<i>-0.0005(2)</i>	<i>0</i>
<b>Rb(1)</b>	<b>0.0236(4)</b>	<b>0.0233(4)</b>	<b>0.0263(4)</b>	<b>0.0017(3)</b>	<b>-0.0012(3)</b>	<b>0.0011(3)</b>
	<i>0.0284(3)</i>	<i>0.0216(3)</i>	<i>0.0302(3)</i>	<i>0.0017(2)</i>	<i>-0.0012(2)</i>	<i>0.0010(2)</i>
<b>Rb(2)</b>	<b>0.0242(6)</b>	<b>0.0666(9)</b>	<b>0.0239(6)</b>	<b>0</b>	<b>-0.0042(4)</b>	<b>0</b>
	<i>0.0287(4)</i>	<i>0.0649(6)</i>	<i>0.0265(4)</i>	<i>0</i>	<i>-0.0046(3)</i>	<i>0</i>
<b>S(1)</b>	<b>0.025(1)</b>	<b>0.032(1)</b>	<b>0.034(1)</b>	<b>0.0117(9)</b>	<b>0.0015(9)</b>	<b>0.0098(8)</b>
	<i>0.0291(6)</i>	<i>0.0312(7)</i>	<i>0.0387(7)</i>	<i>0.0145(6)</i>	<i>0.0009(5)</i>	<i>0.0094(5)</i>
<b>S(2)</b>	<b>0.013(1)</b>	<b>0.031(1)</b>	<b>0.021(1)</b>	<b>0</b>	<b>-0.005(1)</b>	<b>0</b>
	<i>0.0183(8)</i>	<i>0.0305(9)</i>	<i>0.0253(8)</i>	<i>0</i>	<i>-0.0037(6)</i>	<i>0</i>
<b>S(3)</b>	<b>0.024(1)</b>	<b>0.025(1)</b>	<b>0.014(1)</b>	<b>0</b>	<b>0.001(1)</b>	<b>0</b>
	<i>0.0272(8)</i>	<i>0.0259(9)</i>	<i>0.0152(7)</i>	<i>0</i>	<i>0.0000(6)</i>	<i>0</i>

<sup>a</sup> Boldface type refers to  $\text{Rb}_3\text{TaS}_4$ ; italic type refers to  $\text{Rb}_3\text{NbS}_4$ .

methods (11) showed that the  $Fdd2$  space group is the correct one. For the  $\text{Na}_3\text{TaS}_4$  crystal some reflections could not be indexed with the resulting orientation matrix. They were indexed independently with the same unit cell, but a matrix rotated  $90^\circ$  around the  $c$  axis. This rotation superimposes the crystallographic  $a$  and  $b$  axes, which show a difference in length of about 4%. No untwinned crystal was obtained. Since the overall intensity of the second individuum is as low as 10% compared with the main individuum, no twin refinement was attempted. Still it was

possible to determine clearly the sense of the polar axis in this particular crystal. Because the structure refinements on the crystals for both sodium compounds led to the opposite enantiomers, the original and the transformed data for  $\text{Na}_3\text{TaS}_4$  and the original data for  $\text{Na}_3\text{NbS}_4$  are reported in Table 2. All structure refinements were performed using SHELXL-93 (12).

The crystals of the rubidium containing compounds were treated in the same manner as that mentioned for the sodium sulfides. The unit cell parameters and extinction

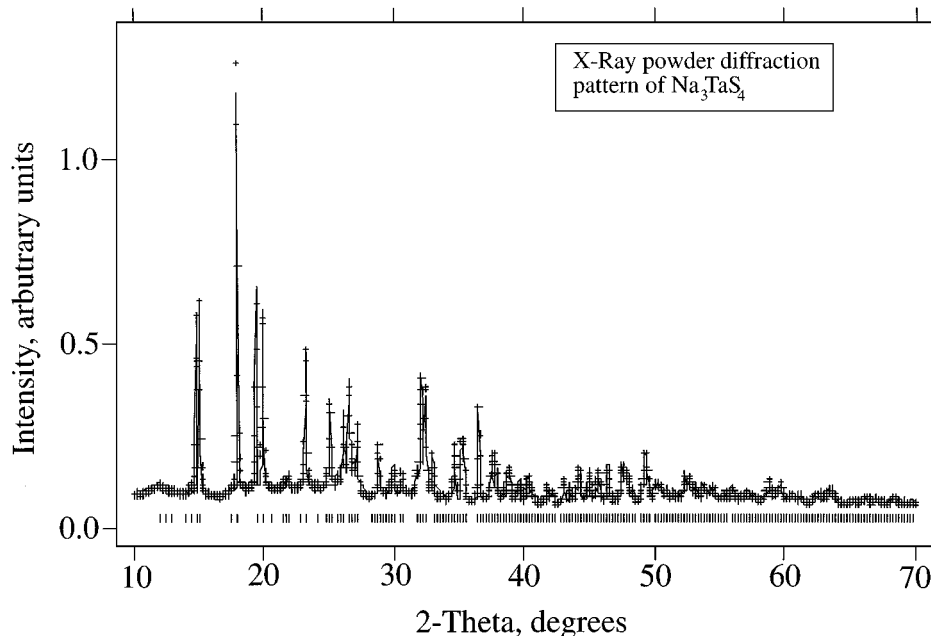
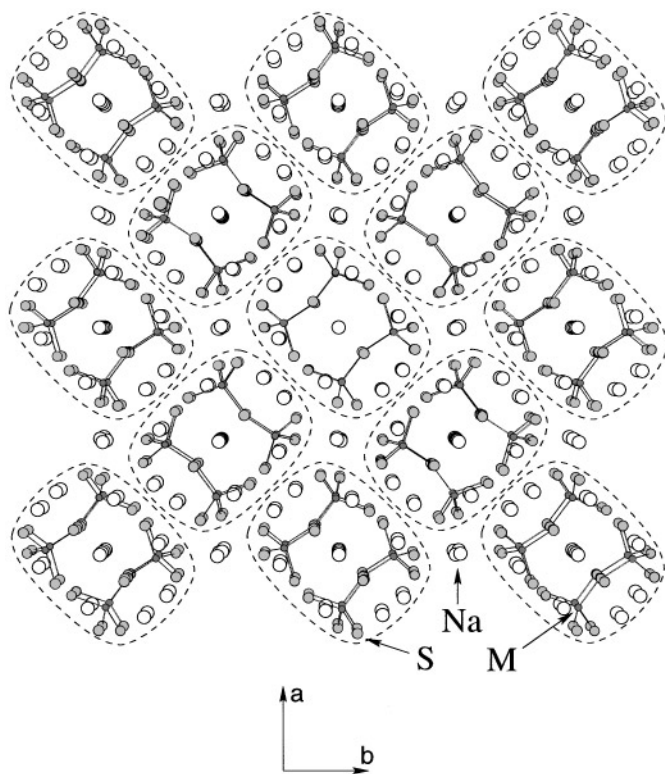


FIG. 1. Powder X-ray diffraction pattern for  $\text{Na}_3\text{TaS}_4$  (crosses), fit based on the single-crystal X-ray data (solid line), and the positions of the peaks (marks at the bottom of the plot).

conditions of data sets on yellow ( $\text{Rb}_3\text{NbS}_4$ ) and orange ( $\text{Rb}_3\text{TaS}_4$ ) crystals led to the space group  $Pnma$ .

Additional information on the structure solution is given in Table 1, fractional atomic coordinates are listed in Tables 2 and 4, and the thermal displacement parameters in Tables 3 and 5.

Both supposedly polycrystalline samples of  $\text{Na}_3\text{TaS}_4$  prepared from stoichiometric components yielded essentially identical powder X-ray diffraction patterns (Scintag XDS 2000 diffractometer with  $\text{CuK}\alpha$  radiation). One is shown in Fig. 1. Also shown are the positions of reflections expected for  $\text{Na}_3\text{TaS}_4$  and the theoretical powder X-ray pattern based on the single crystal data. The positional and thermal displacement parameters of the atoms could not be refined based on the powder X-ray diffraction data (program GSAS (13) was employed), although the match of the experimentally observed diffraction pattern and the one expected based on the single-crystal structure solution is visually flawless. The orthorhombic unit cell parameters of the powders were refined to the following values:  $a = 27.259(2)$  Å,  $b = 28.400(2)$  Å, and  $c = 7.9647(4)$  Å, which agree with the single-crystal data within reasonable estimates of potential errors.



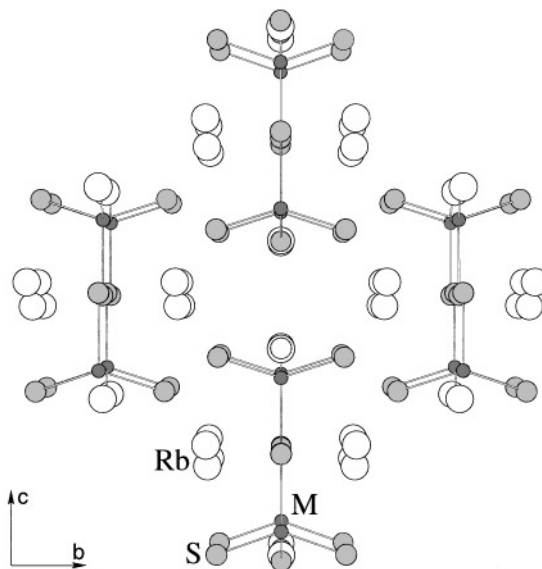
**FIG. 2.** The  $\text{Na}_3\text{MS}_4$  ( $M = \text{Nb}, \text{Ta}$ ) crystal structure viewed along the  $c$  axis. One-dimensional columns of  $\text{MS}_4$  tetrahedra surrounding chains of sodium ions are emphasized. Despite the appearance, the  $\text{MS}_4$  tetrahedra are not directly connected to each other.

## STRUCTURE DESCRIPTIONS AND DISCUSSION

$\text{Na}_3\text{NbS}_4$  and  $\text{Na}_3\text{TaS}_4$  crystallize in a new structure type. A view in this crystal structure is drawn in Fig. 2. It can be described very simply as consisting of isolated  $\text{MS}_4^{3-}$  units ( $M = \text{Nb}, \text{Ta}$ ) separated by sodium cations. However, the packing arrangement in this crystal is quite complex. Figure 2 visually separates the structure into one-dimensional columns running along the  $c$  axis. This special arrangement of the  $\text{MS}_4^{3-}$  units leads to the large  $a$  and  $b$  axes.

$\text{Rb}_3\text{NbS}_4$  and  $\text{Rb}_3\text{TaS}_4$  crystallize isotypically with compounds such as  $\text{K}_3\text{NbS}_4$  (2),  $\text{K}_3\text{VS}_4$  (14), and  $\text{Cs}_3\text{MSe}_4$  ( $M = \text{Nb}, \text{Ta}$ ) (15). Analogous to the sodium compounds the structures are built up from stacking of isolated  $\text{MS}_4^{3-}$  units together with alkali metal cations, but due to the larger size of the latter the stacking occurs in a different manner than that in the sodium compounds. The structure of  $\text{Rb}_3\text{MS}_4$ , viewed along the  $a$  axis, shows chains of  $\text{MS}_4^{3-}$  tetrahedra similar to that in the channels of  $\text{Na}_3\text{MS}_4$  (Fig. 3).

Bond valence sums (16) and Madelung potentials using the Ewald method (17) were calculated for all four title compounds with the program Eutax (18). They are summarized in the Table 6a, b. Important  $M-S$  and  $A-S$  distances, as well as the  $S-M-S$  angles, for all four compounds are listed in the Tables 7a, b and 8a, b. The coordination of the transition metal atoms is nearly tetrahedral, as evidenced by the values of the  $S-M-S$  angles, which range from  $107.6$  to  $112.4^\circ$ . As expected, the  $M-S$  bond lengths ( $2.26$ – $2.29$  Å) show no significant differences between niobium and tantalum. They are characteristic of double bonds. For



**FIG. 3.** Crystal structure of  $\text{Rb}_3\text{MS}_4$  ( $M = \text{Nb}, \text{Ta}$ ) viewed along the  $a$  axis. Rows of  $\text{MS}_4^{3-}$  tetrahedra, similar to that in  $\text{Na}_3\text{MS}_4$  (Fig. 2), are visible. The tetrahedra are not directly connected to each other.

**TABLE 6a**  
Bond Valence Sums and Madelung Potentials (in V) for  $\text{Na}_3\text{MS}_4$   
( $M = \text{Nb, Ta}$ )<sup>a</sup>

Atom	Bond valence sum	Charge	Potential (V)
<b>Ta(1)</b>	<b>5.592</b>	<b>5.00</b>	<b>-39.185</b>
<i>Nb(1)</i>	<i>5.167</i>		<i>-39.004</i>
<b>Ta(2)</b>	<b>5.420</b>	<b>5.00</b>	<b>-38.906</b>
<i>Nb(2)</i>	<i>5.087</i>		<i>-38.885</i>
<b>Na(1)</b>	<b>0.734</b>	<b>1.00</b>	<b>-9.982</b>
	<i>0.737</i>		<i>-10.055</i>
<b>Na(2)</b>	<b>0.759</b>	<b>1.00</b>	<b>-9.458</b>
	<i>0.712</i>		<i>-9.452</i>
<b>Na(3)</b>	<b>1.158</b>	<b>1.00</b>	<b>-11.862</b>
	<i>1.140</i>		<i>-11.817</i>
<b>Na(4)</b>	<b>0.993</b>	<b>1.00</b>	<b>-11.897</b>
	<i>0.979</i>		<i>-11.932</i>
<b>Na(5)</b>	<b>1.110</b>	<b>1.00</b>	<b>-12.020</b>
	<i>1.110</i>		<i>-12.016</i>
<b>Na(6)</b>	<b>0.882</b>	<b>1.00</b>	<b>-7.887</b>
	<i>0.866</i>		<i>-7.793</i>
<b>Na(7)</b>	<b>0.915</b>	<b>1.00</b>	<b>-8.270</b>
	<i>0.903</i>		<i>-8.202</i>
<b>S(1)</b>	<b>2.143</b>	<b>-2.00</b>	<b>19.143</b>
	<i>2.035</i>		<i>19.089</i>
<b>S(2)</b>	<b>2.122</b>	<b>-2.00</b>	<b>20.587</b>
	<i>1.997</i>		<i>20.475</i>
<b>S(3)</b>	<b>2.000</b>	<b>-2.00</b>	<b>19.025</b>
	<i>1.907</i>		<i>18.947</i>
<b>S(4)</b>	<b>2.008</b>	<b>-2.00</b>	<b>20.238</b>
	<i>1.933</i>		<i>20.251</i>
<b>S(5)</b>	<b>2.178</b>	<b>-2.00</b>	<b>21.119</b>
	<i>2.061</i>		<i>21.090</i>
<b>S(6)</b>	<b>2.164</b>	<b>-2.00</b>	<b>19.026</b>
	<i>2.061</i>		<i>18.985</i>
<b>S(7)</b>	<b>1.908</b>	<b>-2.00</b>	<b>18.719</b>
	<i>1.818</i>		<i>18.654</i>
<b>S(8)</b>	<b>2.142</b>	<b>-2.00</b>	<b>21.091</b>
	<i>2.006</i>		<i>21.018</i>

<sup>a</sup> Boldface type refers to  $\text{Na}_3\text{TaS}_4$ , and italic type refers to  $\text{Na}_3\text{NbS}_4$ .

comparison,  $\text{K}_4\text{Ta}_2\text{S}_{11}$  (1) (with  $\text{Ta}^{\text{V}}$ ) features two sets of Ta–S bond lengths: average of 2.235 Å for formally double bonds and a range of 2.465–2.579 Å for formally single bonds. Similar Ta–S bond lengths are found in a complex  $\text{Ta}_6\text{S}_{17}^{4-}$  polyanion in  $(\text{Et}_4\text{N})_4[\text{Ta}_6\text{S}_{17}] \cdot 3\text{CH}_3\text{CN}$  (19): 2.179–2.204 Å for double bonds and 2.400–2.593 Å for single bonds. In the latter compound there are also longer Ta–S contacts of 2.615–2.992 Å.  $\text{CsTaS}_3$  (3) features Ta–S bond lengths of 2.339 and 2.582 Å, in the single bond range. In  $\text{K}_3\text{NbS}_4$  the average bond length is 2.250 Å.  $\text{Cu}_3\text{TaS}_4$  (4),  $\text{Tl}_3\text{TaS}_4$  (5), and  $\text{K}_3\text{TaS}_4$  (2) all contain  $\text{TaS}_4^{3-}$  tetrahedra, but the atomic positions in these compounds are not well established.

The sodium atoms in the  $\text{Na}_3\text{MS}_4$  compounds are coordinated by five (Na(1) and Na(2)) or six (Na(3)–Na(7)) sulfur atoms. The Na(6) and Na(7) are surrounded by dis-

**TABLE 6b**  
Bond Valence Sums and Madelung Potentials (in V) for  $\text{Rb}_3\text{MS}_4$   
( $M = \text{Nb, Ta}$ )<sup>a</sup>

Atom	Bond valence sum	Charge	Potential (V)
<b>Ta</b>	<b>5.520</b>	<b>5.00</b>	<b>-39.242</b>
<i>Nb</i>	<i>5.208</i>		<i>-39.212</i>
<b>Rb(1)</b>	<b>0.949</b>	<b>1.00</b>	<b>-8.214</b>
	<i>0.952</i>		<i>-8.215</i>
<b>Rb(2)</b>	<b>0.733</b>	<b>1.00</b>	<b>-9.007</b>
	<i>0.734</i>		<i>-9.005</i>
<b>S(1)</b>	<b>1.941</b>	<b>2.00</b>	<b>18.970</b>
	<i>1.866</i>		<i>18.967</i>
<b>S(2)</b>	<b>1.959</b>	<b>2.00</b>	<b>19.861</b>
	<i>1.899</i>		<i>19.893</i>
<b>S(3)</b>	<b>2.310</b>	<b>2.00</b>	<b>19.594</b>
	<i>2.215</i>		<i>19.516</i>

<sup>a</sup> Boldface type refers to  $\text{Rb}_3\text{TaS}_4$ ; italic type refers to  $\text{Rb}_3\text{NbS}_4$ .

torted octahedra of sulfur; other polyhedra are rather irregular. The Na–S bond lengths range from 2.79 to 3.26 Å. These are comparable to the Na–S distances in  $\text{Na}_2\text{S}$  (20) (2.82 Å) and in  $\text{Na}_2\text{S}_2$  (21) (2.81 and 2.99 Å). The sum of Shannon (22) ionic radii for  $\text{Na}^+$  (1.02 Å) and  $\text{S}^{2-}$  (1.84 Å) is 2.86 Å. The sodium atoms Na(1) and Na(2) exhibit relatively large thermal parameters (compare Tables 2 and 3), which are mainly the result of large thermal motion in the  $c$  direction along the columns. Both these sodium atoms are five-

**TABLE 7a**  
Ta–S and Na–S Bond Lengths (Å) and S–Ta–S Angles (deg)  
in  $\text{Na}_3\text{TaS}_4$

Ta(1)–S(5)	2.262(6)	Na(3)–S(6)	2.81(1)	Na(6)–S(5) 2 ×	2.88(2)
Ta(1)–S(1)	2.262(7)	Na(3)–S(2)	2.83(1)	Na(6)–S(8) 2 ×	2.93(2)
Ta(1)–S(2)	2.269(7)	Na(3)–S(5)	2.86(1)	Na(6)–S(2) 2 ×	3.252(8)
Ta(1)–S(6)	2.276(7)	Na(3)–S(4)	2.90(1)		
		Na(3)–S(7)	2.98(1)	Na(7)–S(8) 2 ×	2.87(1)
Ta(2)–S(8)	2.269(7)	Na(3)–S(3)	2.99(2)	Na(7)–S(5) 2 ×	2.92(2)
Ta(2)–S(7)	2.276(7)			Na(7)–S(4) 2 ×	3.230(9)
Ta(2)–S(3)	2.277(8)	Na(4)–S(1)	2.79(1)		
Ta(2)–S(4)	2.287(6)	Na(4)–S(6)	2.80(1)	S(1)–Ta(1)–S(5)	108.9(3)
		Na(4)–S(3)	2.89(2)	S(1)–Ta(1)–S(6)	111.3(2)
Na(1)–S(8)	2.88(2)	Na(4)–S(7)	3.02(2)	S(2)–Ta(1)–S(5)	108.9(3)
Na(1)–S(4)	2.92(2)	Na(4)–S(3)	3.20(2)	S(2)–Ta(1)–S(6)	108.1(3)
Na(1)–S(1)	2.97(2)	Na(4)–S(7)	3.21(2)	S(5)–Ta(1)–S(6)	108.9(3)
Na(1)–S(2)	3.16(2)			S(1)–Ta(1)–S(2)	110.8(3)
Na(1)–S(3)	3.23(2)	Na(5)–S(1)	2.83(1)		
		Na(5)–S(4)	2.87(1)	S(3)–Ta(2)–S(4)	112.4(3)
Na(2)–S(5)	2.91(2)	Na(5)–S(7)	2.91(1)	S(3)–Ta(2)–S(7)	108.9(3)
Na(2)–S(2)	2.95(2)	Na(5)–S(2)	2.91(1)	S(3)–Ta(2)–S(8)	109.2(3)
Na(2)–S(3)	2.99(2)	Na(5)–S(6)	2.94(1)	S(4)–Ta(2)–S(7)	107.4(3)
Na(2)–S(6)	3.09(2)	Na(5)–S(8)	2.97(1)	S(4)–Ta(2)–S(8)	108.1(3)
Na(2)–S(1)	3.19(2)			S(7)–Ta(2)–S(8)	110.8(3)
Na–Na ≥	3.52(2)			Na–Ta ≥	3.54(4)

**TABLE 7b**  
**Nb–S and Na–S Bond Lengths (Å) and S–Nb–S Angles (deg)**  
**in Na<sub>3</sub>NbS<sub>4</sub>**

Nb(1)–S(6)	2.277(1)	Na(3)–S(6)	2.831(2)	Na(6)–S(5) 2 ×	2.896(2)
Nb(1)–S(5)	2.273(1)	Na(3)–S(5)	2.843(2)	Na(6)–S(8) 2 ×	2.928(2)
Nb(1)–S(1)	2.264(1)	Na(3)–S(2)	2.844(2)	Na(6)–S(2) 2 ×	3.263(1)
Nb(1)–S(2)	2.287(1)	Na(3)–S(4)	2.919(2)		
		Na(3)–S(3)	2.967(2)	Na(7)–S(8) 2 ×	2.885(2)
Nb(2)–S(8)	2.275(1)	Na(3)–S(7)	2.995(2)	Na(7)–S(5) 2 ×	2.906(2)
Nb(2)–S(7)	2.282(1)			Na(7)–S(4) 2 ×	3.251(1)
Nb(2)–S(3)	2.282(1)	Na(4)–S(6)	2.810(2)		
Nb(2)–S(4)	2.286(1)	Na(4)–S(1)	2.796(2)	S(1)–Nb(1)–S(2)	110.77(4)
		Na(4)–S(3)	2.914(2)	S(1)–Nb(1)–S(5)	108.73(4)
Na(1)–S(8)	2.889(2)	Na(4)–S(7)	2.993(2)	S(1)–Nb(1)–S(6)	111.26(4)
Na(1)–S(4)	2.911(2)	Na(4)–S(3)	3.188(2)	S(2)–Nb(1)–S(5)	109.40(4)
Na(1)–S(1)	2.986(3)	Na(4)–S(7)	3.213(2)	S(2)–Nb(1)–S(6)	107.65(4)
Na(1)–S(2)	3.134(3)			S(5)–Nb(1)–S(6)	108.99(4)
Na(1)–S(3)	3.237(3)	Na(5)–S(1)	2.841(2)		
		Na(5)–S(4)	2.865(2)	S(3)–Nb(2)–S(4)	112.10(4)
Na(2)–S(5)	2.936(2)	Na(5)–S(2)	2.887(2)	S(3)–Nb(2)–S(7)	108.75(4)
Na(2)–S(2)	2.948(2)	Na(5)–S(7)	2.919(2)	S(3)–Nb(2)–S(8)	109.09(4)
Na(2)–S(3)	3.010(3)	Na(5)–S(6)	2.953(2)	S(4)–Nb(2)–S(7)	107.71(4)
Na(2)–S(6)	3.104(2)	Na(5)–S(8)	2.979(2)	S(4)–Nb(2)–S(8)	108.13(4)
Na(2)–S(1)	3.170(3)			S(7)–Nb(2)–S(8)	111.08(4)
Na–Na ≥	3.553(3)			Na–Nb ≥	3.544(2)

fold coordinated by sulfur atoms in a distorted rectangular pyramid. The thermal vibration ellipsoid points directly through the base of these polyhedra with the next sulfur atoms in a distance of more than 3.46 Å. These sodium atoms are also the ones with the lowest bond valence sums (compare Table 6a). In a similar manner the Rb(2) atoms in the rubidium sulfides show a large thermal parameter (see Tables 4 and 5) and exhibit the smallest bond valence sums (Table 6b). The rubidium ions are all sevenfold coordinated by sulfur at distances in the range from 3.28 to 3.91 Å. Comparable distances are found in Rb<sub>2</sub>S (3.313 Å) (23) and Rb<sub>2</sub>S<sub>3</sub> (3.306 to 3.454 Å) (24).

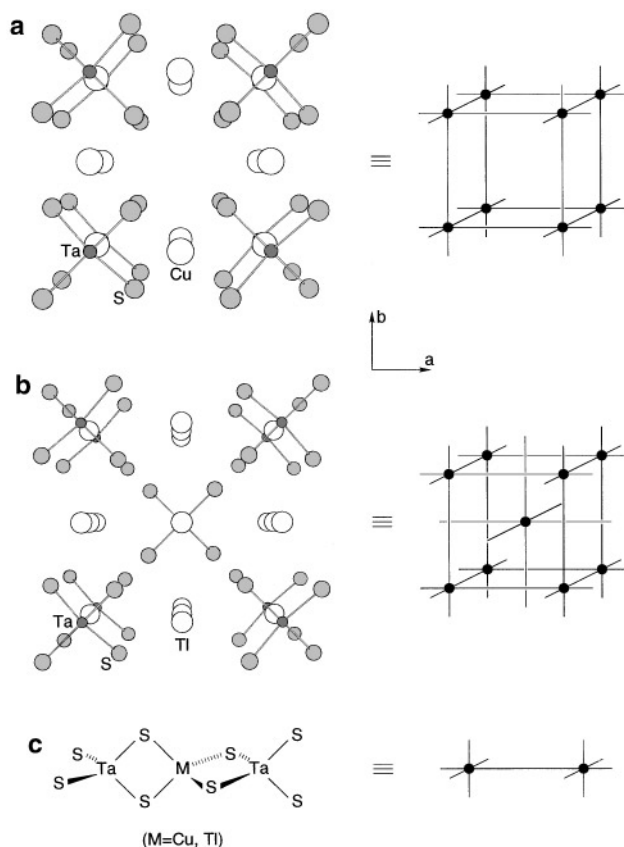
**TABLE 8a**  
**Ta–S and Rb–S Bond Lengths (Å) and S–Ta–S Angles (deg)**  
**in Rb<sub>3</sub>TaS<sub>4</sub>**

Ta(1)–S(3)	2.267(3)	Rb(2)–S(3)	3.279(3)
Ta(1)–S(1) 2 ×	2.268(2)	Rb(2)–S(2)	3.401(3)
Ta(1)–S(2)	2.282(3)	Rb(2)–S(2)	3.550(3)
		Rb(2)–S(1) 2 ×	3.563(3)
Rb(1)–S(3)	3.341(2)	Rb(2)–S(1) 2 ×	3.906(2)
Rb(1)–S(3)	3.353(2)		
Rb(1)–S(1)	3.397(2)	S(1)–Ta(1)–S(1)	112.0(1)
Rb(1)–S(1)	3.400(2)	S(1)–Ta(1)–S(2) 2 ×	109.44(6)
Rb(1)–S(1)	3.509(2)	S(1)–Ta(1)–S(3) 2 ×	108.53(7)
Rb(1)–S(2)	3.529(1)	S(2)–Ta(1)–S(3)	108.83(9)
Rb(1)–S(2)	3.643(2)		
		Rb–Rb ≥	3.919(1)
		Rb–Ta ≥	3.9390(9)

**TABLE 8b**  
**Nb–S and Rb–S Bond Lengths (Å) and S–Nb–S Angles (deg)**  
**in Rb<sub>3</sub>NbS<sub>4</sub>**

Nb(1)–S(3)	2.275(2)	Rb(2)–S(3)	3.277(2)
Nb(1)–S(1) 2 ×	2.269(1)	Rb(2)–S(2)	3.408(2)
Nb(1)–S(2)	2.280(2)	Rb(2)–S(2)	3.544(2)
		Rb(2)–S(1) 2 ×	3.563(2)
Rb(1)–S(3)	3.338(1)	Rb(2)–S(1) 2 ×	3.905(2)
Rb(1)–S(3)	3.353(1)		
Rb(1)–S(1)	3.400(1)	S(1)–Nb(1)–S(1)	112.17(8)
Rb(1)–S(1)	3.399(1)	S(1)–Nb(1)–S(2) 2 ×	109.39(4)
Rb(1)–S(1)	3.513(2)	S(1)–Nb(1)–S(3) 2 ×	108.50(4)
Rb(1)–S(2)	3.529(1)	S(2)–Nb(1)–S(3)	108.83(6)
Rb(1)–S(2)	3.638(1)		
		Rb–Rb ≥	3.921(1)
		Rb–Nb ≥	3.936(1)

It is surprising that the presence of such a simple structural motif as  $MS_4^{3-}$  in  $Na_3MS_4$  results in complex stackings of the anions and alkali metal cations. In trying to rationalize this, comparison with the simpler copper and thallium analogues is instructive. The structures of cubic  $Cu_3TaS_4$  (4) and  $Tl_3TaS_4$  (5) are drawn in Fig. 4. Approximate values for the positional parameters of the sulfur



**FIG. 4.** Crystal structures of cubic  $Cu_3TaS_4$  (a) and  $Tl_3TaS_4$  (b). The schematic drawings on the right are constructed according to the legend (c) at the bottom.

ligands (all metal atoms are in special positions) are used, since a complete single-crystal structure refinement is not available.  $\text{Cu}_3\text{TaS}_4$  can be thought of as a simple cubic array of  $\text{TaS}_4^{3-}$  units connected by  $\text{Cu}^+$  centers. Each  $\text{TaS}_4^{3-}$  moiety is bound to six copper cations via its six edges; each copper center, tetrahedrally coordinated by four sulfur atoms, bridges two  $\text{TaS}_4^{3-}$  tetrahedra.  $\text{Tl}_3\text{TaS}_4$  has a very similar structure: it consists of two interpenetrating cubic  $\text{Cu}_3\text{TaS}_4$ -like networks. This increases the coordination number of thallium atoms from 4 to 8. Although the crystal structures of  $A_3MS_4$  ( $A = \text{Cu, Na, K, Rb, Tl}$ ) are quite different from each other, the following two features are apparent. Firstly, the  $MS_4^{3-}$  tetrahedra are present in all of them; secondly, the packing of  $MS_4^{3-}$  and  $A^+$  correlates fairly with the ionic radii (22) of  $A^+$ :  $\text{Cu}^+$  (0.77 Å) is four-coordinate,  $\text{Na}^+$  (1.02 Å) is five- or six-coordinate,  $\text{K}^+$  (1.38 Å) and  $\text{Rb}^+$  (1.52 Å) are surrounded by seven sulfur centers, and  $\text{Tl}^+$  (1.50 Å) is eight-coordinate.

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